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- [54] **MULTIWAVELENGTH INFRARED FOCAL PLANE ARRAY DETECTOR**
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- [22] Filed: **Jul. 21, 1994**
- [51] **Int. Cl.<sup>6</sup>** ..... **H01L 31/109; H01L 31/0304**
- [52] **U.S. Cl.** ..... **257/190; 257/21; 257/184; 257/440**
- [58] **Field of Search** ..... **257/18, 21, 440, 257/184, 185, 190**

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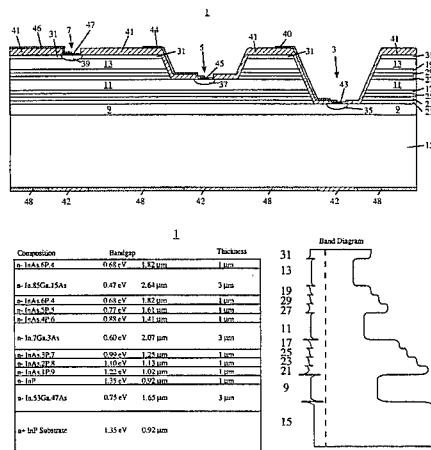
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[57] **ABSTRACT**

A multiwavelength focal plane array infrared detector is included on a common substrate having formed on its top face a plurality of In<sub>x</sub>Ga<sub>1-x</sub>As (x≤0.53) absorption layers, between each pair of which a plurality of InAs<sub>y</sub>P<sub>1-y</sub> (y<1) buffer layers are formed having substantially increasing lattice parameters, respectively, relative to said substrate, for preventing lattice mismatch dislocations from propagating through successive ones of the absorption layers of decreasing bandgap relative to said substrate, whereby a plurality of detectors for detecting different wavelengths of light for a given pixel are provided by removing material above given areas of successive ones of the absorption layers, which areas are doped to form a pn junction with the surrounding unexposed portions of associated absorption layers, respectively, with metal contacts being formed on a portion of each of the exposed areas, and on the bottom of the substrate for facilitating electrical connections thereto.

**9 Claims, 8 Drawing Sheets**



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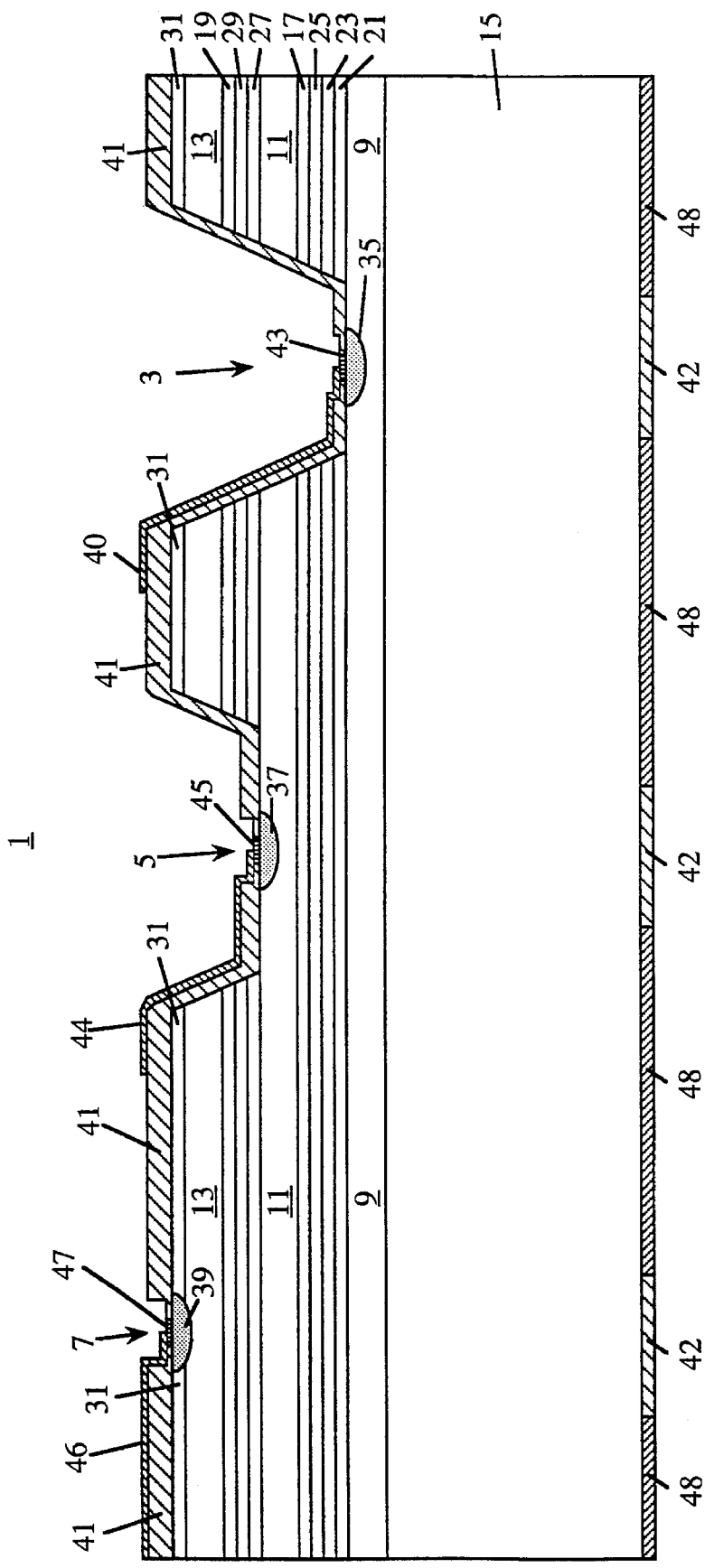


Fig. 1A

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Composition	Bandgap	Thickness
n- InAs.6P.4	0.68 eV	1.82 $\mu\text{m}$
n- In.85Ga.15As	0.47 eV	2.64 $\mu\text{m}$
n- InAs.6P.4	0.68 eV	1.82 $\mu\text{m}$
n- InAs.5P.5	0.77 eV	1.61 $\mu\text{m}$
n- InAs.4P.6	0.88 eV	1.41 $\mu\text{m}$
n- In.7Ga.3As	0.60 eV	2.07 $\mu\text{m}$
n- InAs.3P.7	0.99 eV	1.25 $\mu\text{m}$
n- InAs.2P.8	1.10 eV	1.13 $\mu\text{m}$
n- InAs.1P.9	1.22 eV	1.02 $\mu\text{m}$
n- InP	1.35 eV	0.92 $\mu\text{m}$
n- In.53Ga.47As	0.75 eV	1.65 $\mu\text{m}$
n+ InP Substrate	1.35 eV	0.92 $\mu\text{m}$

Fig. 1B

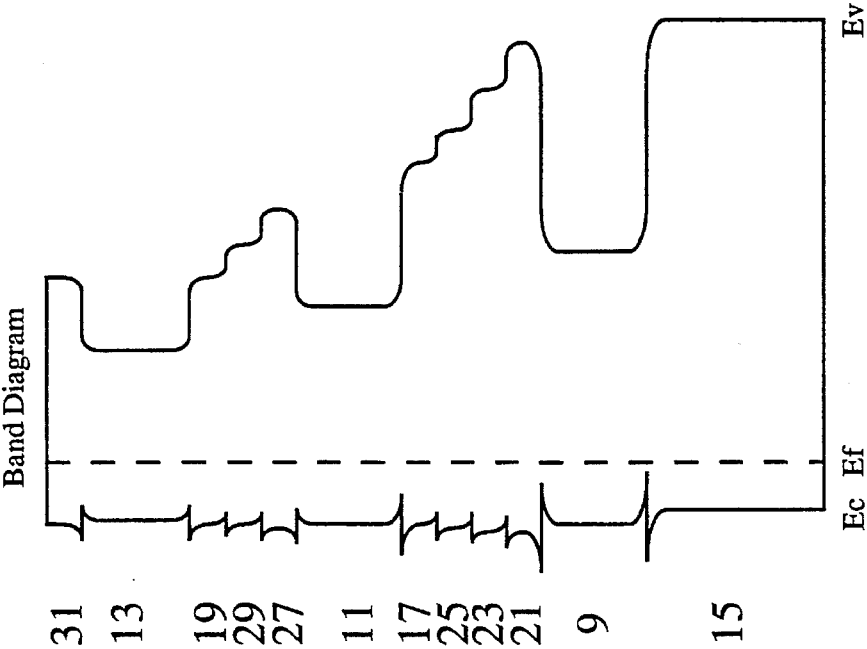


Fig. 1C

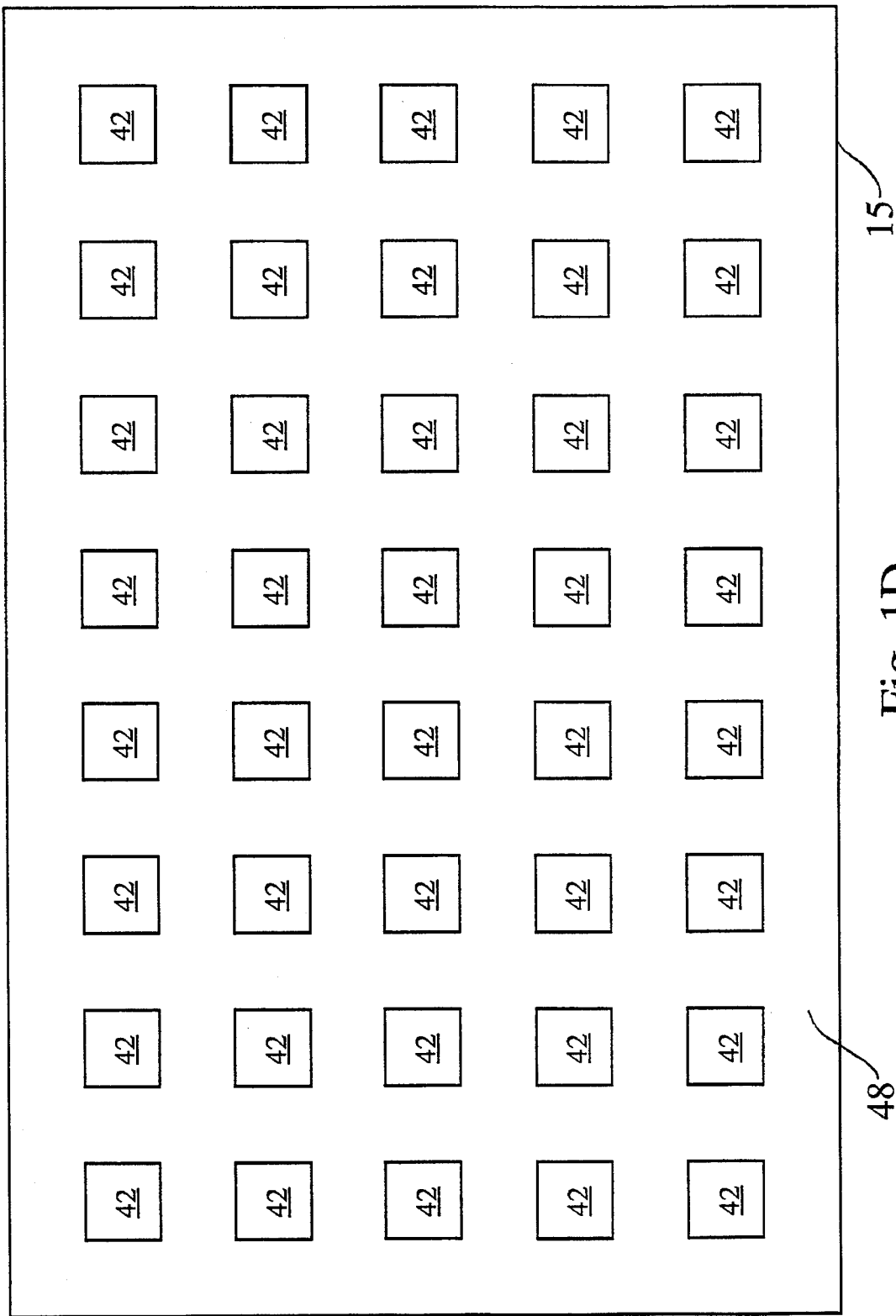


Fig. 1D

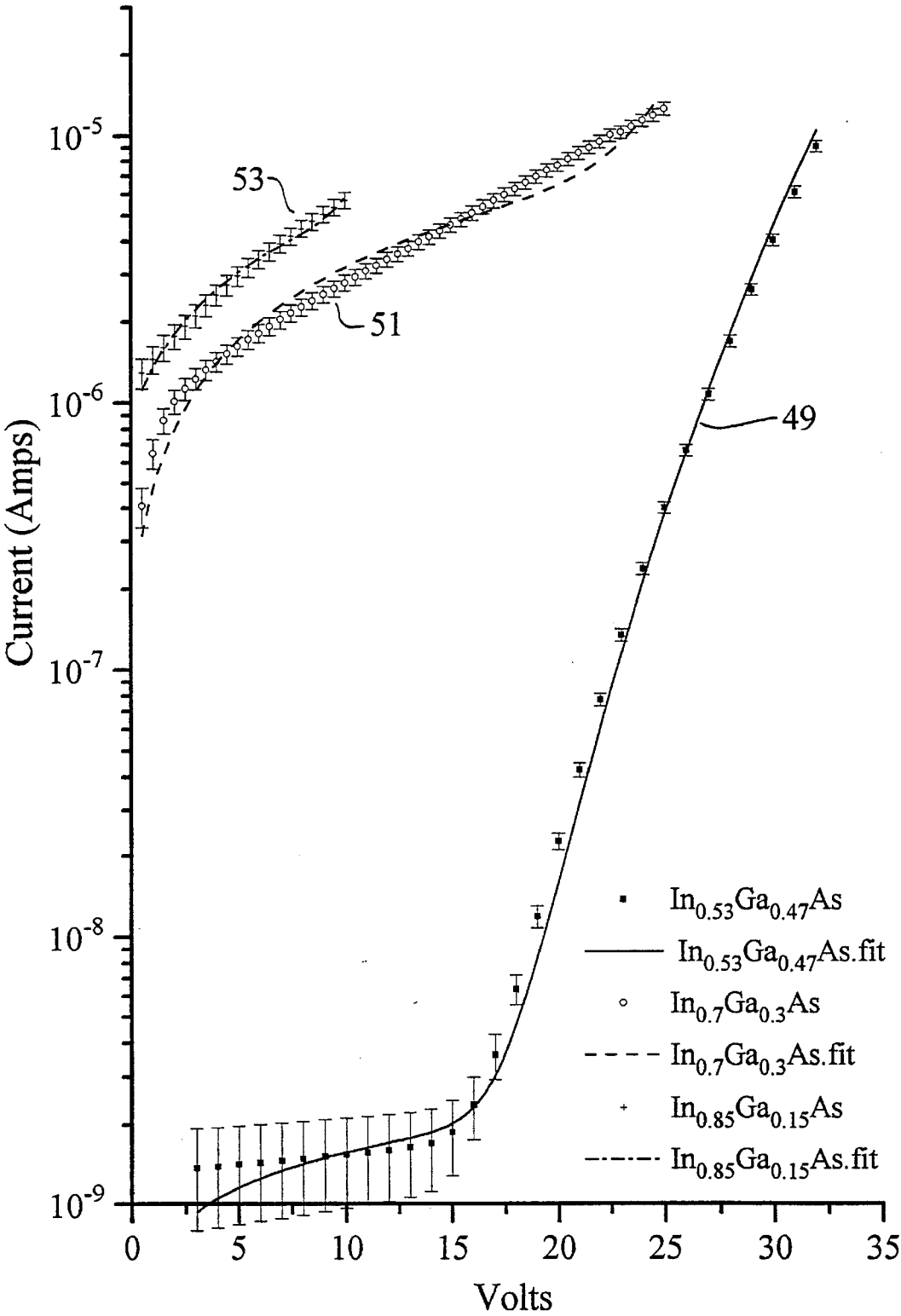


Fig. 2

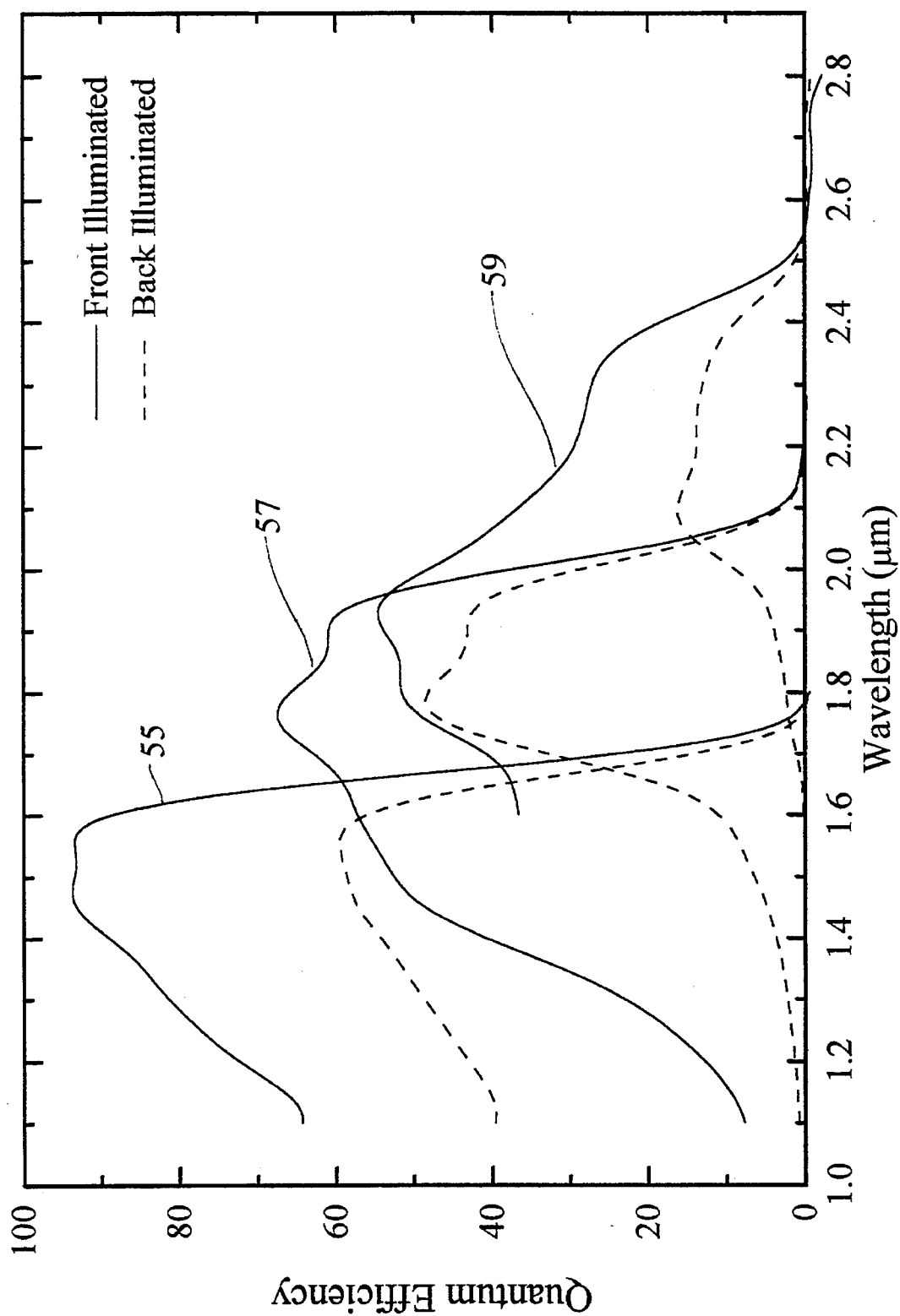


Fig. 3

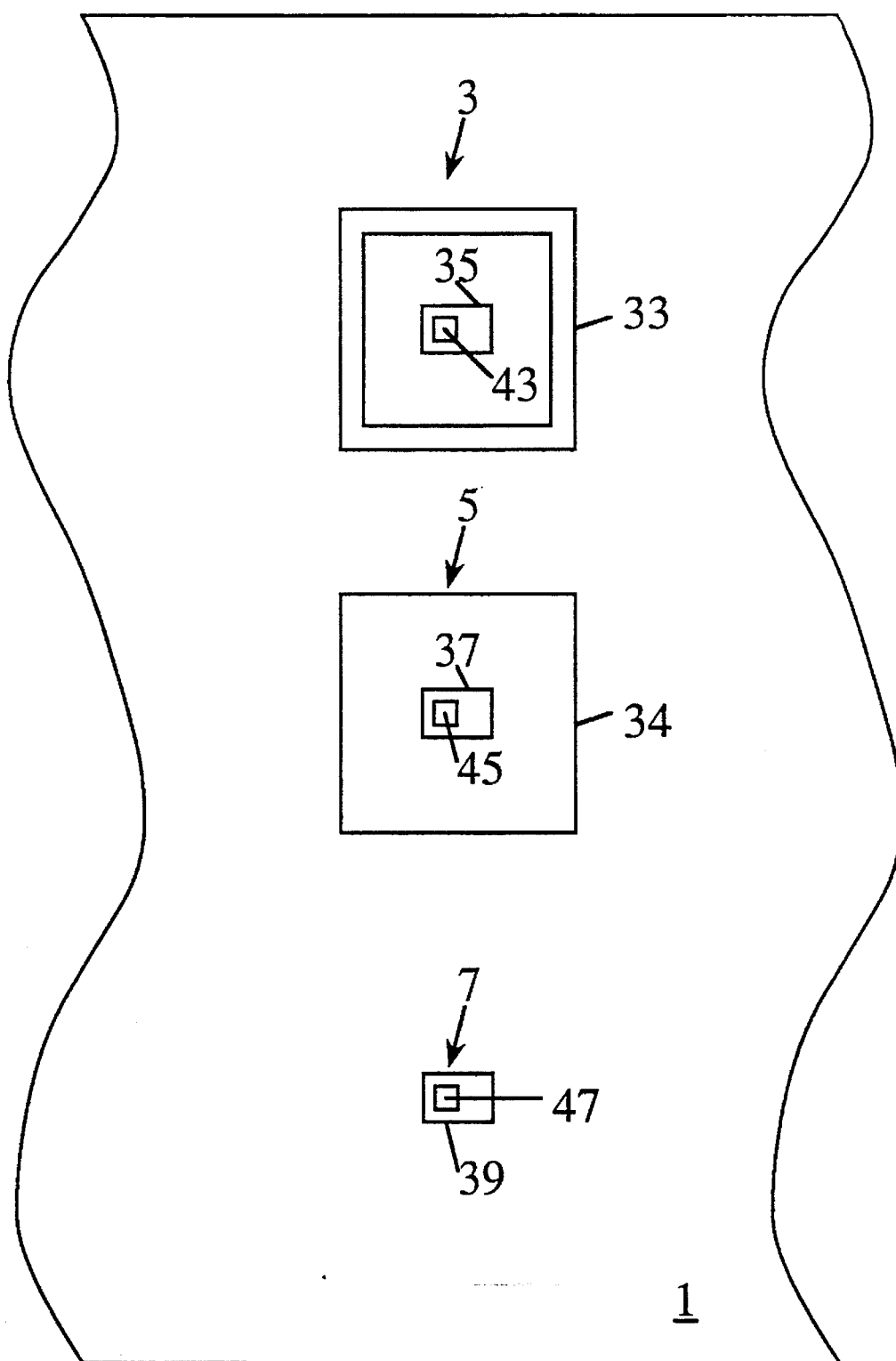


Fig. 4



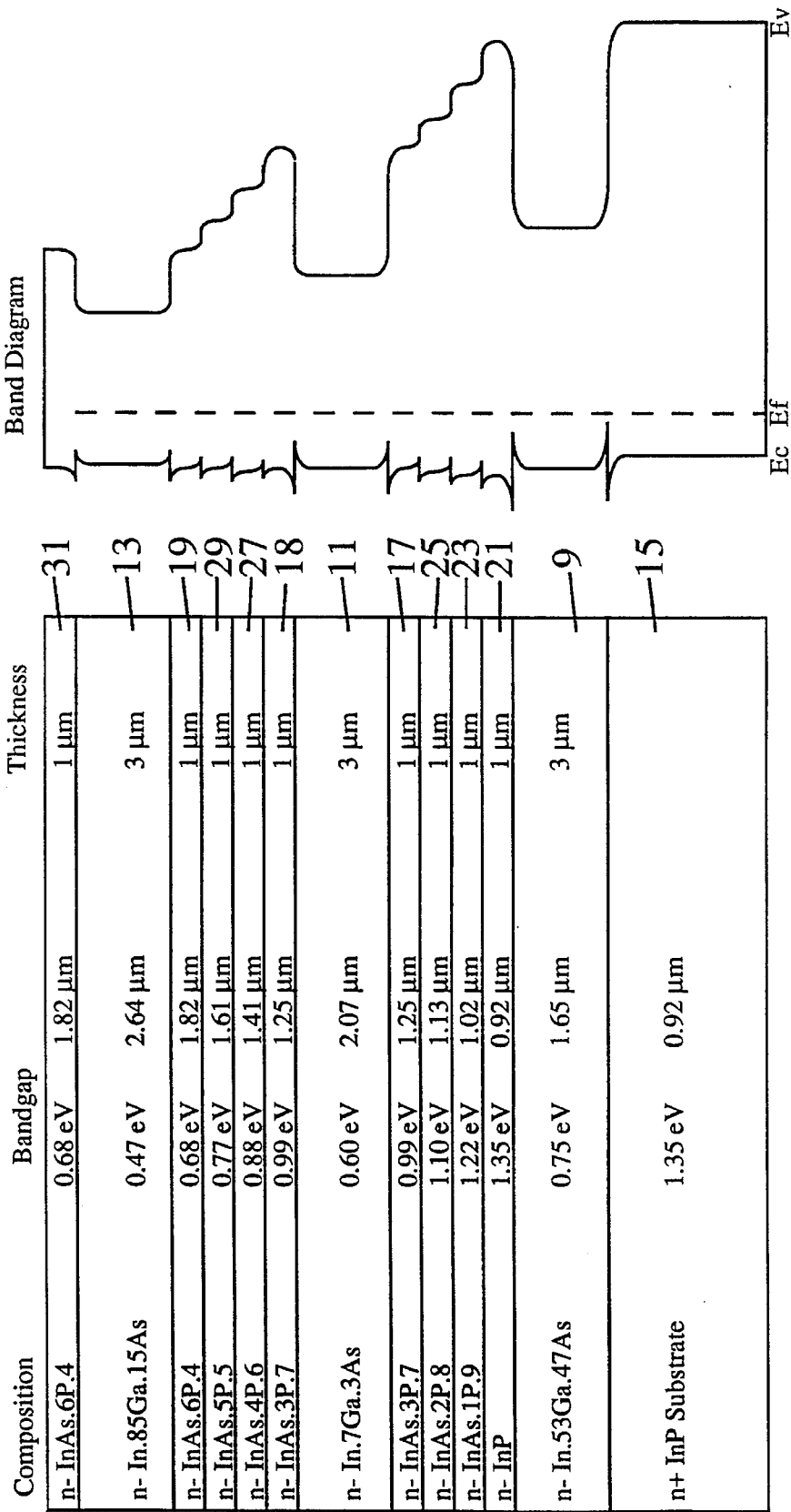


Fig. 5B

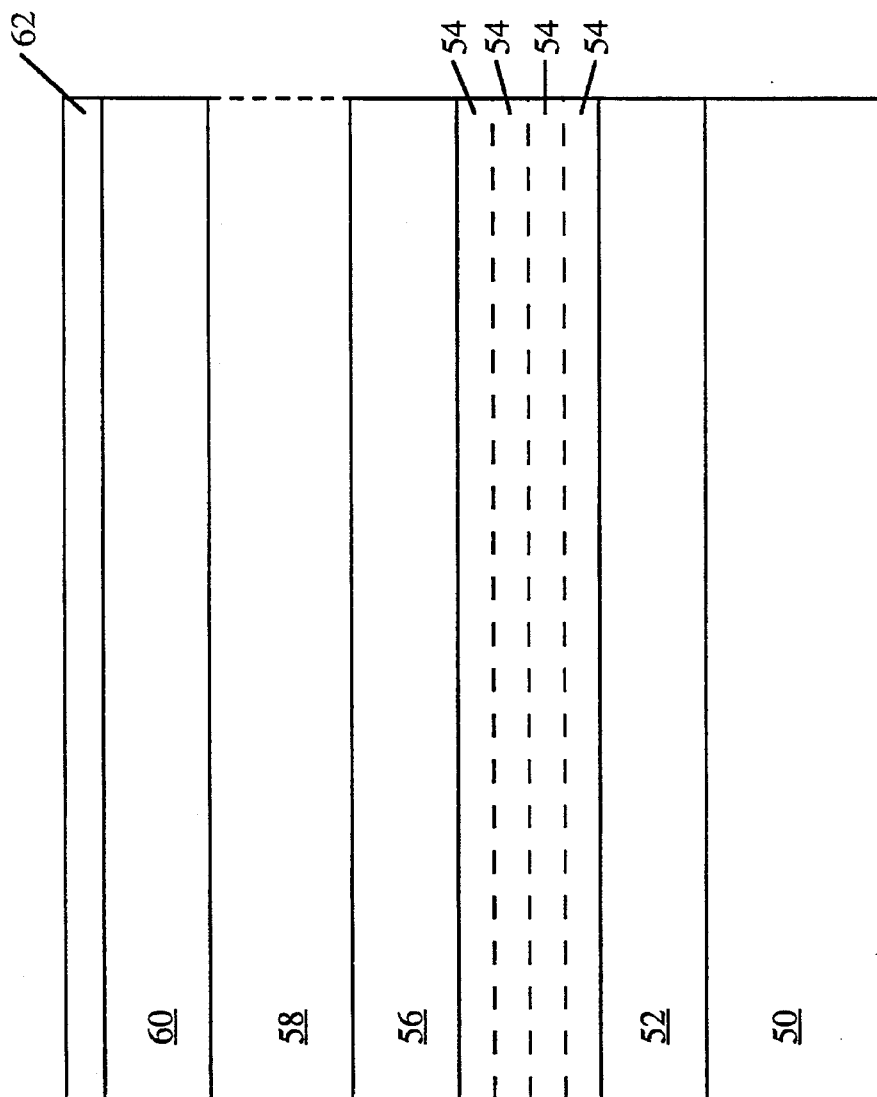


Fig. 6A

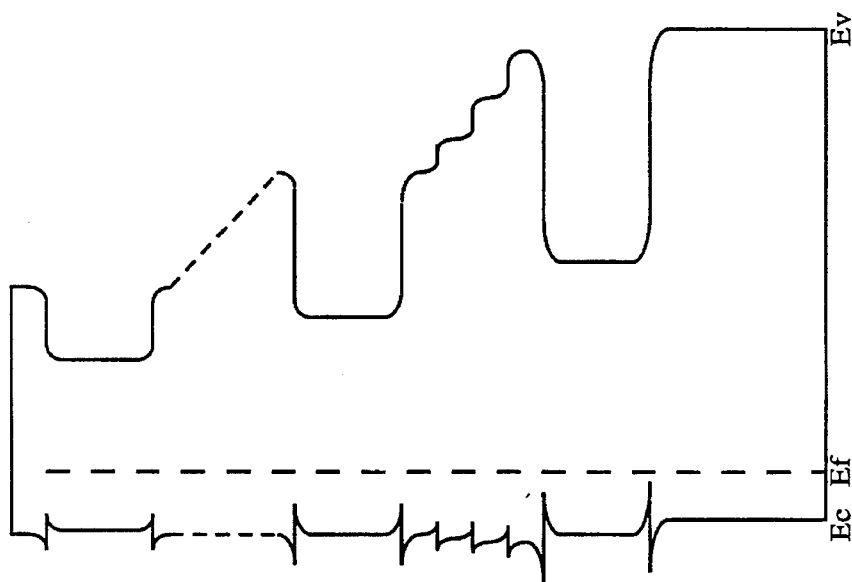


Fig. 6B

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# MULTIWAVELENGTH INFRARED FOCAL PLANE ARRAY DETECTOR

## GOVERNMENT RIGHTS

This invention was made with Government support under Contract No. N00014-93-1-0223P00001, awarded by the Office of Naval Research, Department of Defense, and NASA Jet Propulsion Laboratory, under contract No. NAS7-1304. The Government has certain rights in this invention.

## FIELD OF THE INVENTION

The present invention relates generally to infrared detectors, and more particularly to detectors for detecting light in the infrared of different wavelengths.

## BACKGROUND OF THE INVENTION

Applications involving fiber optic communications systems typically utilize light waves having wavelengths in the near infrared (0.8 to 3.0 micrometers in wavelength). These systems presently represent the greatest usage for near infrared detectors. However, other applications such as temperature sensing, night vision, eye-safe range finding, process control, lidar, and wind-shear detection require detectors with higher sensitivity and faster response times in the near IR region. Recently, InGaAs detectors have been investigated for light detection at wavelengths greater than 1.65  $\mu\text{m}$  because of their potential for high performance and reliability. Such detectors have demonstrated high quantum efficiencies (>70%), low dark current (<100 mA/cm<sup>2</sup> at -5 V), and rise times less than one nanosecond at room temperature. Other materials (Ge, PbS, InSb, PtSi, HgCdTe, etc.) have been used for detectors at wavelengths greater than 2  $\mu\text{m}$ , but they generally have to be cooled to low temperatures, often have very slow response, or have high dark currents.

Since  $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$  detects light at wavelengths  $\leq 1.65 \mu\text{m}$ , in order to detect longer wavelengths more indium must be added to the ternary compound, thereby decreasing the bandgap. In this case, the lattice parameter can no longer match that of the InP substrate. A graded layer technique has been developed to accommodate the lattice mismatch between the substrate ( $a_s = 5.869 \text{ \AA}$ ) and the  $\text{In}_x\text{Ga}_{1-x}\text{As}$  ( $x > 0.53$ ) absorption layer ( $a_a > 5.869 \text{ \AA}$ ). In this technique,  $\text{InAs}_{1-y}\text{P}_y$  ( $y \leq 1$ ) buffer layers with increasing lattice parameters are grown in between the InP substrate and the absorption layer. This prevents lattice mismatch dislocations from propagating from layer to layer, enabling the growth of absorption layers with good optoelectronic properties. It is also possible to use  $\text{In}_x\text{Ga}_{1-x}\text{As}$  as grading layers by increasing the indium concentration. However, by using InAsP which has a larger bandgap than  $\text{In}_x\text{Ga}_{1-x}\text{As}$ , better spectral response is obtained for back illumination (light enters through the substrate). Also, using larger bandgap materials as buffer layers results in detectors with lower dark current. InGaAs detectors with up to 2.6  $\mu\text{m}$  cutoff wavelength using InAsP graded layers have been successfully fabricated with good opto-electronic properties.

## SUMMARY OF THE INVENTION

In one embodiment of the invention, an infrared detector is provided in a plural wavelength InGaAs focal plane array pixel element for detecting a plurality of wavelengths of light over a predetermined range in the infrared or near infrared, where each of the wavelength sensitive detectors are formed on a common substrate, and are individually

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addressable. The detector consists of successively smaller bandgap layers of  $\text{In}_x\text{Ga}_{1-x}\text{As}$  ( $x \geq 0.53$ ) formed over a substrate, separated by compositionally graded layers of  $\text{InAs}_y\text{P}_{1-y}$  ( $y \leq 1$ ) to decrease defects induced by lattice mismatch strain with the InP substrate. Portions of the various layers are selectively removed to form different pn junctions with different wavelength responses, respectively

## BRIEF DESCRIPTION OF THE DRAWINGS

Various embodiments of the present invention are described below with reference to the drawings, in which like items are identified by the same reference designation, and in which:

FIG. 1A is a cross sectional view of an engineering prototype for a three wavelength infrared detector in one embodiment of the invention;

FIG. 1B is a pictorial and chart-like diagram showing for each layer of the device of FIG. 1A, the composition, bandgap, and thickness thereof, respectively;

FIG. 1C is a band diagram positioned in alignment with the corresponding layers of the device of FIG. 1B, for showing the relative bandgaps of each layer compared to the other, respectively;

FIG. 1D shows the bottom of the detector of FIG. 1A, for a preferred embodiment thereof including an antireflective coating and ohmic contact grid thereon;

FIG. 2 shows a plot of the measured dark current of each one of the three detectors of the device of FIG. 1A under various conditions of reverse bias voltage;

FIG. 3 shows a plot of quantum efficiency versus wavelength for each one of the three detector regions of the device of FIG. 1A;

FIG. 4 is a top view of a portion of the device of FIG. 1A, showing three detector regions;

FIG. 5A shows a simplified cross section of the epitaxial structure of an ideal device for another embodiment of the invention;

FIG. 5B shows a band diagram illustrating the band gaps for each one of the corresponding layers of the device of FIG. 5A;

FIG. 6A shows a cross section of the epitaxial structure of an n-color device of another embodiment of the invention; and

FIG. 6B shows a band diagram of the band gaps of each of the layers of the device of FIG. 6A.

## DETAILED DESCRIPTION OF THE INVENTION

In FIG. 1A, a prototype detector array pixel sensitive to three different selectable infrared wavelengths is shown, for one embodiment of the invention as developed for use as a focal plane array imaging device for applications such as gas spectroscopy and absolute temperature measurements (pyrometry). Since detectors with smaller bandgap which detect longer wavelength also have larger dark current, one can use this pixel to select the detector with the appropriate absorption layer to maximize the quantum efficiency while minimizing the dark current. The device 1 includes a plurality of integrated detector pixels, each including regions 3, 5, and 7 for detecting different wavelengths of light, respectively. In this example, the light is on the near infrared or infrared wavelength regions. Optical absorption occurs, with reference to FIG. 1B, in 3  $\mu\text{m}$  thick  $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$ ,

$\text{In}_{0.7}\text{Ga}_{0.3}\text{As}$ , and  $\text{In}_{0.85}\text{Ga}_{0.15}\text{As}$  layers **9**, **11**, **13**, respectively, which are grown by vapor phase epitaxy on top of an InP substrate **15**,  $\text{InAs}_{0.3}\text{P}_{0.7}$  layer **17**, and  $\text{InAs}_{0.6}\text{P}_{0.4}$  layer **19**, respectively. The bandgaps of the absorption layers **9**, **11**, **13** are 0.75, 0.60, and 0.47 eV, respectively, which correspond to cutoff wavelengths of 1.65, 2.07, 2.64  $\mu\text{m}$ , respectively. To accommodate the lattice mismatch between the absorption layers **9**, **11**, and **13**, step-graded 1  $\mu\text{m}$  thick InAsP buffer layers **21**, **23**, **25**, **17**, **27**, **29**, and **19** are grown. Thus, the respective lattice parameters of the  $\text{In}_x\text{Ga}_{1-x}\text{As}$  absorption layers matches the lattice parameter of the InAsP layers immediately underneath. The lattice parameter of the InAsP layers immediately above the absorption layers should also match, but in this structure the lattice parameter of  $\text{InAs}_{0.4}\text{P}_{0.6}$  layer **27** does not exactly match that of  $\text{In}_{0.7}\text{Ga}_{0.3}\text{As}$  absorption layer **11**. This may be the source of somewhat elevated dark currents in the longer wavelength detectors. Accordingly, in a preferred device the lattice parameters of layers **27** and **11** should match. All the grown layers were undoped (with a background n-type carrier concentration of  $<5 \times 10^{15} \text{ cm}^{-3}$ ), while the (100) InP substrate **15** was doped with sulfur to give an n-type doping density of  $8 \times 10^{18} \text{ cm}^{-3}$ . The band diagram in FIG. **1C** shows the band gap and the band offsets between the different layers **15**, **9**, **21**, **23**, **25**, **17**, **11**, **27**, **29**, **19**, **13**, and **31**; and FIG. **1B** shows the composition, bandgap in electron volts (eV) and equivalent wavelength in micrometers ( $\mu\text{m}$ ), and the thickness in micrometers ( $\mu\text{m}$ ), respectively.

A selective wet etching process was developed in order to access the different absorption layers to enable junction diffusion for detection regions **3**, **5**, and **7**, respectively (see FIG. **1A**). A mixture of 5:1 citric acid (50% by weight) :  $\text{H}_2\text{O}_2$  was used to etch InGaAs detection layers since it is strongly selective of InGaAs (versus InAsP), and leaves a good surface morphology. The etch rate of  $\text{In}_x\text{Ga}_{1-x}\text{As}$  is  $\sim 1000 \text{ \AA/min}$  at room temperature. To etch InAsP layers, a mixture of  $\text{HCl}:\text{H}_3\text{PO}_4:\text{H}_2\text{O}_2$  in the ratio of 3:1:x was used, where x was varied from 0 to 0.3 as the arsenic concentration in InAsP was increased. The etch rate of InAsP is  $\sim 200 \text{ \AA/sec}$  at room temperature. To fabricate the integrated detector array **1**, a area square area **33** of  $500 \mu\text{m}^2$  (see FIG. **4**) was first etched above the  $\text{In}_{0.7}\text{Ga}_{0.3}\text{As}$  layer **11** and  $\text{In}_{0.85}\text{Ga}_{0.15}\text{As}$  layer **13**, using a plasma deposited  $\text{SiN}_x$  film (1000  $\text{\AA}$ ) as an etch mask, for initially forming detection region **3**. Similarly, a  $500 \mu\text{m}^2$  area **34** was etched above layer **11** for initially forming detector region **5**. Thin layers (1  $\mu\text{m}$  thick) **21** and **27** of InAsP were left on top of the absorption layers in regions **3** and **5** as a wider bandgap cap layer in order to reduce surface-generated dark current. The pn junctions of all three detectors in regions **3**, **5**, and **7**, respectively, were formed in 100 by 150  $\mu\text{m}$  areas **35**, **37**, and **39**, respectively, using a single sealed ampoule diffusion of Zinc Arsenide, with  $\text{SiN}_x$  used as the diffusion mask. As a result, p+ diffusions were formed in regions **35**, **37**, and **39**, thereby providing pn junctions with their underlying n doped absorption layers **9**, **11**, and **13**, respectively. Next, an antireflection coating **41** of  $\text{SiN}_x$  is deposited to a thickness of 2250  $\text{\AA}$  on the top diode surface (see FIG. **1A**). Also, 40  $\mu\text{m}$  square Au-Zn alloy contacts **43**, **45**, and **47** are placed on top of the diffused areas **35**, **37**, **39**, respectively, using a photoresist lift-off process. Also, overlay metal contacts **40**, **44**, and **46**, typically of TiAu material, are formed on top of the antireflective coating **41** in association with detectors **3**, **5**, and **7**, for electrically contacting contacts **43**, **45**, and **47**, respectively. Contacts **43**, **45**, and **47** facilitate making electrical connections either to individual ones of, or two or more of detectors **3**, **5**, and **7**, respectively, using integrated

circuitry techniques, such as flip-chip bonding, or wire bonding.

In order to maximize performance, the back surface of the detector array **1** or bottom of substrate **15** should preferably have an antireflection coating **42** of  $\text{SiN}_x$ , and an ohmic contact grid **48** (typically GeNiAu alloy), as shown in FIG. **1D**. More specifically, in the preferred embodiment, the contact **48** is formed into a grid pattern with open spaces **42** for allowing backlighting or back illumination of the substrate **15** or detector **1**. The open spaces **42** consist of a transparent antireflective coating of  $\text{SiN}_x$ , in this example. Each space **42** permits back illumination of an underlying absorption layer **9**, **11**, or **13**, associated with a given detector **3**, **5**, or **7**, of a pixel in an array of such pixels, in this example.

The basic processing steps for the three wavelength infrared focal plane array detector element **1** of FIG. **1A** are summarized in eight steps as follows:

- I. Deposit 1.000  $\text{\AA}$   $\text{SiN}_x$  film by Plasma Enhanced Chemical Vapor Deposition (PECVD)
- II. Photolithography to define etch area (500  $\mu\text{m}$  square)
- III. Material selective wet etching
  - InGaAs: Citric Acid: $\text{H}_2\text{O}_2$  at 5:1
  - InAsP:  $\text{HCl}:\text{H}_3\text{PO}_4:\text{H}_2\text{O}_2$  at 3:1:x
- IV. Deposit 1000  $\text{\AA}$   $\text{SiN}_x$  diffusion mask
- V. Photolithography to define diffusion area (100 by 150  $\mu\text{m}$  area)
- VI. Sealed ampoule diffusion using  $\text{Zn}_2\text{As}_3$  at 500° C.–530° C. for 20–40 minutes
- VII. Deposit 2250  $\text{\AA}$   $\text{SiN}_x$  anti-reflective coating on top surface
- VIII. Place Au-Zn alloy contacts (40  $\mu\text{m}$  square) on top of diffused area using photoresist lift-off process
- IX. Deposit  $\text{SiN}_x$  AR coating on substrate surface
- X. Deposit GeNiAu on substrate surface over previously exposed and developed photoresist layer lift off metal to form grid pattern

Note that in step II, a chrome or iron oxide photolithography mask can be used. In step III, in place of wet etching, reactive ion (dry) etching can be used. Also, in step IV, a silicon nitride mask is used.

Using capacitance versus voltage measurements, the inventors obtained the carrier concentration in the absorption layers **9**, **11**, and **13** for each detector **3**, **5**, and **7**, respectively. It was determined that the background carrier concentration in the absorption layers **11** and **13** is  $<1 \times 10^{16} \text{ cm}^{-3}$ , but higher for the  $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$  layer **9** where the carrier concentration increases near the heavily doped substrate **15**, due to the diffusion of the sulfur substrate dopant into the epitaxially grown layer. Also, at 0 V, the smaller bandgap materials have higher capacitance (2.1, 3.0, 7.8 pF for  $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$ ,  $\text{In}_{0.7}\text{Ga}_{0.3}\text{As}$ , and  $\text{In}_{0.85}\text{Ga}_{0.15}\text{As}$  layers **9**, **11**, and **13**, respectively). In the operating range of 5–10 V, all diodes or pn junctions exhibit capacitances ranging from 1.2–2.0 pF, again with the short wavelength ( $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$ ) detector **3** having the smallest capacitance.

The dark current of each detector **3**, **5**, and **7** under reverse bias is shown as points in FIG. **2**, in plots **49**, **51**, and **53**, for absorption layers **9**, **11**, and **13**, respectively. The error bars were determined from the sum of measurement random error (determined to be five percent of the measured value) and small systematic errors. The lines are theoretical fits assuming that the total dark current is the sum of the generation-recombination current (either in the bulk or at the surface), junction shunt current, and diffusion current at low

voltages, while tunneling dominates at high voltages, as has been shown to be the case in previous studies of InGaAs photodiodes.

The theory fits the measured data, especially for the In<sub>0.53</sub>Ga<sub>0.47</sub>As layer 9 for detector 3. The main source of low voltage dark current for this detector is generation-recombination current which is (at  $V > kT$ ) given by:

$$I_{gr} = \frac{qn_i A W}{\tau_{eff}} \quad (1)$$

where  $k$  is the Boltzman constant,  $T$  is the absolute temperature,  $q$  is the electronic charge,  $\tau_{eff}$  is the effective carrier lifetime,  $n_i$  is the intrinsic carrier concentration,  $A$  is the surface or cross-sectional area of the depletion region boundary, and  $W$  is the depletion region width for an abrupt one-sided junction. From the fit, the value of  $\tau_{eff}$  is estimated to be 1  $\mu$ s, indicating that the growth and processing of the complex structure shown in FIG. 1 does not significantly affect the diode properties. The tunneling current, which becomes dominant at  $V > 15$  volts for this detector 3, in this example, is given by:

$$I_{tun} \equiv \gamma A \exp \left( \frac{\Theta m_0^{1/2} \epsilon_g^{3/2}}{q \eta E_m} \right) \quad (2)$$

where  $m_0$  is the free electron mass,  $\epsilon_g$  is the energy band gap of the absorbing layer material,  $\eta$  is Planck's constant divided by  $2\pi$ ,  $E_m$  is the maximum junction electric field given by:

$$E_m = -2(V + V_{bi})/W \quad (3)$$

and  $\Theta$  depends on the shape of the tunneling barrier. Here,  $\Theta$  was estimated to be 0.26 from the fit. The prefactor  $\gamma$  depends on the initial and final states of the tunneling carrier.

The dark currents of In<sub>0.7</sub>Ga<sub>0.3</sub>As and In<sub>0.85</sub>Ga<sub>0.15</sub>As detectors 5 and 7, respectively, are considerably larger than for the In<sub>0.53</sub>Ga<sub>0.47</sub>As detector 3, especially at lower voltages. This is due, in part, to the smaller bandgap of the former materials which not only leads to an increased intrinsic carrier concentration affecting both the diffusion and the generation-recombination currents, but also leads to increased tunneling current. Another source of the high dark current is the larger concentration of defects in these materials caused by the lattice mismatch between the absorption layers 11 and 13, and the InP substrate 15. These defects provide midgap generation-recombination centers, increasing the generation-recombination current. Indeed,  $\tau_{eff}$  for the In<sub>0.7</sub>Ga<sub>0.3</sub>As layer 11 is estimated to be 110 ns, which is nearly an order of magnitude less than for In<sub>0.53</sub>Ga<sub>0.47</sub>As layer 9.

The a.c. small signal conductance at 0 V was measured at 1 kHz to be 18.2 nS, 4.54  $\mu$ S, 9.34  $\mu$ S which translates to shunt resistances of 55.1 M $\Omega$ , 220 k $\Omega$ , and 107 k $\Omega$  for In<sub>0.53</sub>Ga<sub>0.47</sub>As layer 9, In<sub>0.7</sub>Ga<sub>0.3</sub>As layer 11, and In<sub>0.85</sub>Ga<sub>0.15</sub>As layer 13 for detectors 3, 5, and 7, respectively. Assuming that the generation-recombination is the main source of conductance near 0 V, one can calculate  $\tau_{eff}$  (using these conductance values) for In<sub>0.53</sub>Ga<sub>0.47</sub>As and In<sub>0.7</sub>Ga<sub>0.3</sub>As layers 9, 11, respectively, to be 1.1  $\mu$ s and 61 ns, respectively, which is in good agreement with values calculated from the dark current.

The contribution from shunt current, given by:

$$I_{ohm} = V/R_{eff} \quad (4)$$

where  $R_{eff}$  is the effective resistance, was found to be much

greater for In<sub>0.7</sub>Ga<sub>0.3</sub>As and In<sub>0.85</sub>Ga<sub>0.15</sub>As detectors, where  $R_{eff}$  was approximately 3–4 M $\Omega$ , while for In<sub>0.53</sub>Ga<sub>0.47</sub>As detector,  $R_{eff} > 5$  G $\Omega$ . This may also be due to the larger number of defects in the In<sub>0.7</sub>Ga<sub>0.3</sub>As and In<sub>0.85</sub>Ga<sub>0.15</sub>As layers, but the physical origin of the shunt conduction is not clear.

The diffusion current is negligible except for the In<sub>0.85</sub>Ga<sub>0.15</sub>As detector. The diffusion current is given by:

$$i_{diff} = qn_i^2 \sqrt{\frac{D_p A}{\tau_d N_d}} \quad (5)$$

where  $D_p$  is hole diffusion constant,  $\tau_d$  is the minority carrier diffusion lifetime, and  $N_d$  is the doping density. The diffusion current depends exponentially on the bandgap which is smallest for the In<sub>0.85</sub>Ga<sub>0.15</sub>As layer 13, where  $\tau_d$  was estimated to be 500 ps.

Note that the tunneling current contribution to the dark current was not observed for the In<sub>0.7</sub>Ga<sub>0.3</sub>As and In<sub>0.85</sub>Ga<sub>0.15</sub>As detectors 5 and 7 due to the large component of generation, diffusion, and shunt currents. It is believed that the integration and processing of the three-detector pixel in the above example for detector array 1 does not significantly degrade individual device performance.

The quantum efficiency of each detector 3, 5, and 7 under front (light incident from the associated pn junction) and back illumination is shown in plots 55, 57, and 59, respectively, of FIG. 3. The measurements were made under a reverse bias of 3.5, 6.0, 5.0 volts for In<sub>0.53</sub>Ga<sub>0.47</sub>As, In<sub>0.7</sub>Ga<sub>0.3</sub>As, and In<sub>0.85</sub>Ga<sub>0.15</sub>As detectors 3, 5, and 7, respectively. The measured long wavelength cutoffs of 1.7, 2.1, and 2.5  $\mu$ m correspond to the bandgaps of the absorption layer materials. The short wavelength cutoff for the device 1 under back illumination is determined by the light absorption properties of layers between the substrate 15 and the absorption layers 9, 11, and 13. For example, for the In<sub>0.7</sub>Ga<sub>0.3</sub>As layer forming detector 5, light absorbed in the underlying In<sub>0.53</sub>Ga<sub>0.47</sub>As layer 9 will not be detected, thus the short wavelength cutoff of the In<sub>0.7</sub>Ga<sub>0.3</sub>As detector 5 is approximately equal to the cutoff wavelength of In<sub>0.53</sub>Ga<sub>0.47</sub>As layer 9. The peak quantum efficiency under front illumination ranges from 55 to 95%. For the In<sub>0.85</sub>Ga<sub>0.15</sub>As detector 7, the peak quantum efficiency was determined to be 55%. This lower than expected efficiency is believed due, in part, to the large number of heterojunctions and layers underlying the detectors, increasing the probability of the carrier being captured and recombining at traps prior to being collected. Also, the fact that the diffusion of Zn is faster in In<sub>0.85</sub>Ga<sub>0.15</sub>As layer 13 compared with the other absorption layers 9 and 11, caused the thickness of the depleted absorption region to be less than optimum (<2  $\mu$ m) in this detector 1, affecting its quantum efficiency. The peak quantum efficiency under back illumination (between 15% and 60%) is somewhat lower than for front illumination since the antireflective coating was deposited only on the top surface of device 1, in this particular example.

In summary of one embodiment of the invention, as described above, a novel three wavelength InGaAs focal plane array pixel element 1 for detection at wavelengths from 0.9–2.6  $\mu$ m is shown, where each of three wavelength-sensitive detectors 3, 5, and 7 are individually addressable. This device 1 consists of successively smaller bandgap layers of In<sub>x</sub>Ga<sub>1-x</sub>As ( $x \geq 0.53$ ) 9, 11, and 13, grown on an InP substrate 15, separated by layers of InAs<sub>y</sub>P<sub>1-y</sub> to decrease defects induced by lattice mismatch strain with the substrate 15. The various layers were selectively removed so that pn junctions with different wavelength response can be

separately contacted. All three detectors 3, 5, and 7 have quantum efficiencies between 15 and 95% (depending on wavelength and illumination direction) and dark currents from 0.01 to 10 mA/cm<sup>2</sup>—values comparable to discrete photodiodes with similar wavelength responses.

To improve the performance of the three wavelength infrared focal plane array detector 1 of FIGS. 1A, and 1B, the present inventors believe that the modified device as shown in FIG. 5A is preferred. As shown, relative to the prototype detector element 1 of FIG. 1B, the preferred embodiment thereof of FIG. 5A includes an additional transparent strain relief layer 18 between absorption layer 11 and strain relief layer 27, as shown. Improved performance is expected to be obtained in that the lattice parameter of the In<sub>0.3</sub>P<sub>0.7</sub> layers 17 and 18, immediately below and above the absorption layer 11, are matched. In this manner, the magnitude of the dark current associated with absorption layer 11 is expected to be reduced, as previously indicated above. Also, the improved performance can be observed by comparing the band diagram shown in FIG. 1C for the prototype device 1, relative to the band diagram of FIG. 5B for the preferred configuration of FIG. 5A, whereby as shown the bandgap is extended for strain relief layers 19, 29 and 27, by the addition of strain relief layer 18.

The present invention, within practical limits, can be extended to provide a focal plane array detector element capable of detecting “N” different wavelengths, where N is any integer number 1, 2, 3, 4, 5, . . . N. As shown in FIG. 6A, such a device includes a cap layer 62, analogous to layer 31 of FIG. 5A, a substrate 50, a first absorption layer 52, under strain relief layers 54 (analogous to layers 17, 25, 23, and 21 of the device of FIG. 5A), a second absorption layer 56, followed by alternating buffer absorption layers 58 to the Nth order or degree, followed by an Nth absorption layer, followed by the previously mentioned cap layer 62. The material for each of these layers is generally indicated in Table 1 as shown below, as is the doping for each of these layers. Also, in FIG. 6B a band diagram is included showing the bandgaps of the various layers associated with the N absorption layer device of FIG. 6A.

TABLE 1

Layer	Name	Material	Doping
50	Substrate	InP	n+
52	First Absorption Layer	InGaAs	Undoped (n-)
54	Buffer Layers	InAsP, GaAlAsSb, InGaPSb, InGaAsSb, or InAlAsSb	Undoped (n-)
56	Second Absorption Layer	InGaAs	Undoped (n-)
58	Alternating Buffer and Absorption Layers		Undoped (n-)
60	Nth Absorption Layer	InGaAs	Undoped (n-)
62	Cap Layer	Same as Layer C	Undoped (n-)

Although various embodiments of the invention are described herein for purposes of illustration, they are not meant to be limiting. Those of skill in the art may recognize modifications that can be made in the illustrated embodiments. Such modifications are meant to be covered by the spirit and scope of the appended claims. For example, a plurality of pixels each providing the capability of detecting up to (N+1) different wavelengths of light can be provided on a common substrate, with each pixel including (N+1)

detectors, thereby providing an array of such pixels through use of the present invention.

What is claimed is:

1. A multiwavelength focal plane array light detector, comprising:
  - a substrate doped to have a first conductivity, said substrate having top and bottom faces;
  - a plurality of (N+1) absorption layers formed over the top of said substrate, where N=1,2,3,4, . . . , the composition of each of said plurality of absorption layers being individually predetermined for absorbing or detecting different wavelengths of light, respectively;
  - a plurality of (N-1) groups of light transparent buffer layers sandwiched between each successive pair of said plurality of (N+1) absorption layers, each of said groups of buffer layers including a plurality of buffer layers, said buffer layers of each group having successively decreasing bandgaps the further a buffer layer is from said substrate relative to other buffer layers of its associated said group of buffer layers, and at least all but one buffer layer of each said group of buffer layers having smaller bandgaps than the buffer layers of any other of said groups of said buffer layers closer to said substrate, respectively, relative to said substrate, for substantially preventing lattice mismatch dislocations from propagating through successive ones of said plurality of absorption layers, said buffer layers each having said first conductivity;
  - a plurality of spaced apart holes formed from a top of said detector through selected successive ones of said absorption layers and buffer layers, for exposing predetermined areas of top portions of successive ones of said absorption layers, respectively, each of said exposed areas of said absorption layers being doped to form an underlying region in the associated absorption layer having a second conductivity opposite that of said first conductivity;
  - a plurality of first ohmic contacts individually formed in a portion of the holes and on top of a portion of each of said exposed areas of said absorption layers; and
  - a second ohmic contact formed on the bottom of said substrate, whereby a plurality of light detecting diodes are provided relative to each exposed area of said absorption layers, respectively, for individually detecting light of different wavelengths, respectively.
2. The light detector of claim 1, wherein for each successive second through (1+Nth) absorption layer, the same lattice parameter is provided for the ones of said buffer layers formed on top and bottom surfaces thereof, respectively.
3. The light detector of claim 1, further including:
  - an antireflective coating on top and bottom faces thereof exclusive of said plurality of first ohmic contacts, and said second ohmic contact, respectively.
4. The light detector of claim 1, further including a transparent capping layer of buffer layer material formed over top portions of said detector exclusive of said exposed areas of said absorption layers, respectively.
5. The light detector of claim 4, further including:
  - an antireflective coating on topmost and bottommost faces of said detector exclusive of said plurality of first ohmic contacts, and said second ohmic contact.
6. The light detector of claim 1, wherein said plurality of (N+1) absorption layers consist of In<sub>x</sub>Ga<sub>1-x</sub>As (X≥0.53), whereby beginning with N=1 for first and second absorption layers formed over said substrate, and succeeding (N+1)

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ones of said absorption layers, each have successively decreasing bandgaps relative to said substrate, respectively, for detecting successively increasing or longer wavelengths of light.

7. The light detector of claim 6, wherein said plurality of N buffer layers consist of  $\text{InAs}_y\text{P}_{1-y}$  material where  $y \leq 1$ .

8. The light detector of claim 7, wherein said substrate consists of InP material.

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9. The light detector of claim 8, wherein each one of said plurality of (N+1) absorption layers is sandwiched between two of said buffer layers of substantially the same composition, whereby a first one of said absorption layers formed directly over said InP substrate, is covered by a buffer layer of InP material substantially the same as that of said substrate.

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